

## Vanadium oxide ( $V_xO_y$ ) thin films obtained by pulsed laser deposition Películas delgadas de óxidos de vanadio ( $V_xO_y$ ) obtenidas por depósito por láser pulsado

L. Escobar-Alarcón <sup>a,\*</sup>, D. Fuentes-Molina <sup>b</sup>, L.A. Martínez-Chávez <sup>b</sup>, K. Esquivel <sup>b</sup>, D.A. Solis-Casados <sup>c</sup>

<sup>a</sup> Departamento de Física, Instituto Nacional de Investigaciones Nucleares, Carretera México-Toluca S/N, la Marquesa, Ocoyoacac, Estado de México. C.P.52750 México.

<sup>b</sup> División de Investigación y Posgrado, Facultad de Ingeniería, Universidad Autónoma de Querétaro, Cerro de las Campanas, Santiago de Querétaro, Querétaro 76010, México.

<sup>c</sup> Universidad Autónoma del Estado de México, Facultad de Química, Centro Conjunto de Investigación en Química Sustentable UAEM-UNAM, México.

### Abstract

Vanadium oxide thin films were successfully deposited onto glass substrates using the pulsed laser deposition (PLD) technique from a high-purity vanadium target. Structural and physicochemical characterizations by Raman and X-ray photoelectron spectroscopy (XPS) confirmed the coexistence of vanadium pentoxide ( $V_2O_5$ ), vanadium dioxide ( $VO_2$ ), possibly vanadium oxide nanotubes, and sodium vanadate ( $NaVO_3$ ). Scanning electron microscopy revealed microneedles, microbars, and nanotube-like morphologies, while UV-Vis diffuse reflectance spectroscopy indicated both direct and indirect optical transitions, with band gap values influenced by the coexistence of mixed phases. Photoluminescence spectra displayed emission lines attributable to  $V_2O_5$ . These results highlight the versatility of PLD for producing multifunctional vanadium oxide films with mixed oxidation states and controlled morphology, avoiding chemical precursors and providing a sustainable route toward applications in photocatalysis, smart windows, solar cells, and optoelectronic devices.

**Keywords** Vanadium Oxides, Pulsed Laser Deposition, Thin Films.

### Resumen

Se depositaron con éxito películas delgadas de óxido de vanadio sobre sustratos de vidrio mediante la técnica de depósito por láser pulsado (PLD) a partir de un blanco de vanadio de alta pureza. Las caracterizaciones estructurales y fisicoquímicas mediante espectroscopía fotoelectrónica de rayos X (XPS) y Raman confirmaron la coexistencia de pentóxido de vanadio ( $V_2O_5$ ), dióxido de vanadio ( $VO_2$ ), posiblemente nanotubos de óxido de vanadio y vanadato de sodio ( $NaVO_3$ ). La microscopía electrónica de barrido reveló microagujas, microbarras y morfologías similares a nanotubos, mientras que la espectroscopía de reflectancia difusa UV-Vis indicó transiciones ópticas tanto directas como indirectas, con valores de banda prohibida influenciados por la coexistencia de mezclas de fases. Los espectros de fotoluminiscencia mostraron líneas de emisión atribuibles al  $V_2O_5$ . Estos resultados resaltan la versatilidad de PLD para producir películas multifuncionales de óxido de vanadio con estados de oxidación mezclados y morfología controlada, evitando precursores químicos y proporcionando una ruta sostenible hacia aplicaciones en fotocatalisis, ventanas inteligentes, celdas solares y dispositivos optoelectrónicos.

**Palabras Clave:** Óxidos de Vanadio, Depósito por Láser Pulsado, Películas Delgadas.

### 1. Introduction

Vanadium is a transition metal, discovered by a Spanish-Mexican scientist in Zimapán, Hidalgo, Mexico in 1801. This metal with multiple oxidation states, forms a wide variety of vanadium oxides such as  $V_2O_5$ ,  $VO_2$ ,  $V_2O_3$ , and  $VO$  corresponding to the +5, +4, +3 and +2 oxidation states respectively. It can also form oxides that combine oxidation states, e.g.  $V_6O_{13}$  ( $V^{5+}/V^{4+}$ ),  $V_8O_{15}$ ,  $V_7O_{13}$ ,  $V_6O_{11}$  ( $V^{4+}/V^{3+}$ )

(Barceloux and Barceloux, 1999). Thus, vanadium oxides constitute an important family of materials with diverse crystal structures and physicochemical properties (Liu et al., 2007). These materials exhibit metallic, semiconducting, and insulating behavior, enabling applications in a variety of technological areas (Hu et al., 2023), such as photocatalysis (Poossekheaw et al., 2022), water treatment (Gonzalez-Zavala et al., 2016), Li-ion batteries (Mai et al., 2011 Julien et al 2001), and solar cells (Chen et al., 2019).

\*Autor para la correspondencia: [luis.escobar@inin.gob.mx](mailto:luis.escobar@inin.gob.mx)

**Correo electrónico:** [dfuentes12@alumnos.uaq.mx](mailto:dfuentes12@alumnos.uaq.mx) (David Fuentes Molina), [luis.alejandro.martinez@uaq.edu.mx](mailto:luis.alejandro.martinez@uaq.edu.mx) (Luis Alejandro Martínez Chávez), [karen.esquivel@uaq.mx](mailto:karen.esquivel@uaq.mx) (Karen Esquivel Escalante), [solis\\_casados@yahoo.com.mx](mailto:solis_casados@yahoo.com.mx) (Dora Alicia Solis Casados), [luis.escobar@inin.gob.mx](mailto:luis.escobar@inin.gob.mx) (Luis Escobar Alarcón)

**Historial del manuscrito:** recibido el 05/09/2025, última versión-revisada recibida el 31/10/2025, aceptado el 05/11/2025, publicado el 12/12/2025. DOI: <https://doi.org/10.29057/icbi.v13iEspecial4.12548>

Vanadium dioxide ( $\text{VO}_2$ ) and vanadium pentoxide ( $\text{V}_2\text{O}_5$ ), two thermodynamically stable phases, with band gaps of  $\sim 0.6$  and  $\sim 2.3$  eV, respectively, have been widely studied as chromogenic materials because their optical properties respond to external stimuli such as electromagnetic radiation, electric charge, or heat, which makes them suitable for optoelectronics and solar cells for energy saving windows applications (Bahlawane and Lenoble, 2014). Among chromogenic materials, thermochromic materials constitute a subcategory that changes their properties in response to temperature variations. This response is passive, as it is driven by ambient temperature rather than user intervention (Aburas *et al.*, 2019).

$\text{VO}_2$  has been proposed as a thermochromic material owing to its relatively very low transition temperature ( $T_c$ ) of  $68^\circ\text{C}$ , from semiconductor to metal behavior, which enables modulation of incident light and reduces infrared transmittance (Wang *et al.*, 2016). Numerous studies have lowered the  $T_c$  by doping  $\text{VO}_2$  with elements such as molybdenum (Zhang *et al.*, 2012), tungsten (Salamati *et al.*, 2019), and boron (Zhou *et al.*, 2020), achieving a minimum  $T_c$  of  $28.1^\circ\text{C}$ . In contrast,  $\text{V}_2\text{O}_5$  exhibits highly anisotropic electrical and optical properties. The  $T_c$  of crystalline  $\text{V}_2\text{O}_5$  thin films occurs at approximately  $257^\circ\text{C}$  (Aita *et al.*, 1986). The monocrystalline  $\text{V}_2\text{O}_5$  (001) surface in thin film form can undergo a reversible transition between  $77$ – $127^\circ\text{C}$  that is confined to the surface layer; bulk  $\text{V}_2\text{O}_5$  does not show this behavior (Blum *et al.*, 2007). Consequently,  $\text{V}_2\text{O}_5$  thin films are promising for various applications, including optical and electrical switching, thermochromic devices, and thermal sensors (Bannuru *et al.*, 2008).

Among thin film formation methods, pulsed laser deposition (PLD) is a powerful and versatile technique for preparing thin films with tailored properties which can have a wide range of potential applications. In this line, our group has been working in the implementation of several experimental configurations to develop thin films for photocatalysis under simulated solar irradiation (Gonzalez-Zavala, *et al.*, 2016; Escobar-Alarcón *et al.*, 2024), Li-ion micro-batteries (Julien *et al.*, 1999), high-hardness coatings (Escobar-Alarcón *et al.*, 2005), and surface enhanced Raman spectroscopy (Morales Méndez *et al.*, 2023), among others. In this work, we present results on the deposition and characterization of vanadium oxide thin films prepared by PLD. Thin films were characterized by several techniques to know their physical and chemical properties. X-ray photoelectron spectroscopy (XPS) was used to determine the elemental chemical composition and the oxidation state of the elements present; UV-Vis spectroscopy was used to determine the band gap energy, the microstructure was characterized by micro-Raman spectroscopy, the surface morphology was examined by scanning electron microscopy (SEM) and Photoluminescence was employed to determine the defects present in the material.

## 2. Experimental

### 2.1. Thin film deposition

Vanadium oxide thin films were deposited by PLD on glass substrates ( $2.5 \times 2.5$  cm). The target was a high-purity vanadium disk (99.99%) with a diameter of 2.54 cm and a thickness of 0.63 cm. It was ablated with a Nd:YAG laser (1064 nm, 5 ns pulse duration) operating at 10 Hz. Depositions were carried out at  $3 \times 10^{-4}$  Torr, with a target-substrate distance of 5 cm. The deposition time was 60 min for all samples. Films were prepared at three laser fluences: 7.3, 13.3 and  $30.6 \text{ J/cm}^2$ . After deposition, the films were subjected to a thermal treatment at  $450^\circ\text{C}$  for 3 hours in an open-atmosphere furnace to completely oxidize any remaining metallic vanadium and to improve crystallinity. The samples exhibited yellowish to reddish-brown colors, consistent with vanadium oxides.

### 2.2. Physicochemical characterization

Raman spectra were acquired with a micro-Raman system equipped with an Olympus BX40 confocal microscope. The 532 nm line of a Nd:YAG laser was used, and two positions were measured on each film. Surface morphology was examined with a JEOL JSM 6510LV scanning electron microscope (SEM). X-ray photoelectron spectroscopy (XPS) was performed with a Thermo Scientific K-alpha XPS spectrometer, to determine surface composition and oxidation states. Photoluminescence (PL) spectra were collected on a Horiba Jobin Yvon FluoroMax-4 spectrofluorometer using a 500 nm excitation. Diffuse reflectance UV-Vis spectra were obtained with a Varian Cary 100 Conc spectrophotometer to estimate optical properties.

## 3. Results and discussion

### 3.1. SEM characterization

SEM images (Figure 1) show that the film deposited at the lowest fluence ( $7.3 \text{ J/cm}^2$ ) presents agglomerated needle-shaped micrometer-scale structures (Figure 1a). A closer view reveals two-dimensional arrangements formed by ordered agglomeration of needles which are aligned, by group, in the same direction (Figure 1b). The sample deposited at  $13.3 \text{ J/cm}^2$  shows individual needles dispersed on the surface with little agglomeration (Figure 1c); higher magnification reveals microbars approximately  $20 \mu\text{m}$  in length (Figure 1d). For the sample prepared at  $30.6 \text{ J/cm}^2$ , in addition to microscale droplets, some needles are observed (Figure 1e), and at a higher magnification nanoribbons or nanotubes are evident (Figure 1f). All three films display needle-like morphologies, suggesting a common vanadium oxide phase, and the possible formation of vanadium oxide nanotubes, consistent with reported elongated, needle-like nanotube morphologies characteristic of vanadium oxide with outer diameters of 15–100 nm (Muhr *et al.*, 2000).

### 3.2. Microstructural characterization

Raman spectra corresponding to the sample prepared at  $7.3 \text{ J/cm}^2$  are very similar in the two different points measured (Figure 2a). Features at 87, 112, 136, 196, 288, 320, 476, and

992  $\text{cm}^{-1}$  are assigned to the vibrational modes of vanadium pentoxide ( $\text{V}_2\text{O}_5$ ) (Liu *et al.*, 2006; Ureña-Begara *et al.*, 2017). Signals at 223, 254, and 430  $\text{cm}^{-1}$  correspond to vanadium dioxide ( $\text{VO}_2$ ) (Schilbe, 2002; Pan *et al.*, 2004; Théry *et al.*, 2016). Bands at 170, 805, 885, and 912  $\text{cm}^{-1}$  are consistent with  $\text{V}_7\text{O}_{16}$  nanotubes (Julien, *et al.*, 1997; Ureña-Begara, *et al.*, 2017). Peaks at 429, 556, 731, 886, 909, and 943  $\text{cm}^{-1}$  indicate  $\text{NaVO}_3$  (Utegulov *et al.*, 2003; Neuville, 2006).

Raman spectra of the 13.3  $\text{J}/\text{cm}^2$  sample are also similar at both measurement positions (Figure 2b).  $\text{V}_2\text{O}_5$  bands appear at 84, 110, 136, 192, 277, 304, and 480  $\text{cm}^{-1}$  (Liu *et al.*, 2006; Ureña-Begara *et al.*, 2017). Signals at 220, 255, 503, 594 and 672  $\text{cm}^{-1}$  are assigned to  $\text{VO}_2$  (Schilbe, 2002; Pan *et al.*, 2004; Théry *et al.*, 2016); while features at 160, 255, and 809  $\text{cm}^{-1}$  are consistent with  $\text{V}_7\text{O}_{16}$  nanotubes (Julien, *et al.*, 1997;

Ureña-Begara, *et al.*, 2017). Low intensity peaks at 429 and 731  $\text{cm}^{-1}$  can be attributed to  $\text{NaVO}_3$  (Utegulov *et al.*, 2003; Neuville, 2006). Different Raman spectra were obtained for the film prepared at 30.6  $\text{J}/\text{cm}^2$ , indicating a non-uniform phase distribution (Figure 2c).  $\text{V}_2\text{O}_5$  signals are located at 107, 137, 198, 282, 510, and 992  $\text{cm}^{-1}$  (Liu *et al.*, 2006; Ureña-Begara, *et al.*, 2017).  $\text{VO}_2$  bands occur at 262, 344, 430, 603, and 668  $\text{cm}^{-1}$  (Schilbe, 2002; Pan *et al.*, 2004; Théry *et al.*, 2016). Vibrational modes at 170, 885, 916, and 940  $\text{cm}^{-1}$  may arise from  $\text{V}_7\text{O}_{16}$  nanotubes (Julien *et al.*, 1997; Ureña-Begara *et al.*, 2017). Low intensity peaks at 430, 635, 886, 916, and 954  $\text{cm}^{-1}$  are attributed to  $\text{NaVO}_3$  (Utegulov *et al.*, 2003; Neuville, 2006). Overall, Raman results indicate that the films consist of mixtures of different vanadium-oxide phases.

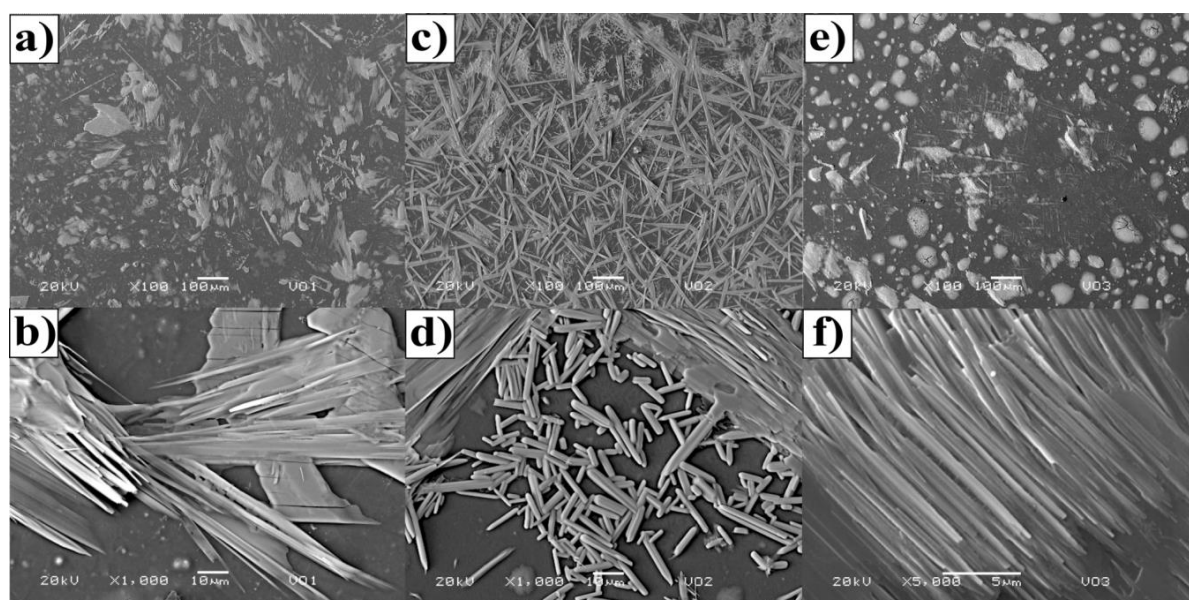


Figure 1. SEM images at 100 $\times$  and 1000 $\times$  of films deposited at laser fluences of 7.3  $\text{J}/\text{cm}^2$  (a,b), 13.3  $\text{J}/\text{cm}^2$  (c,d), and 30.6  $\text{J}/\text{cm}^2$  (e,f).

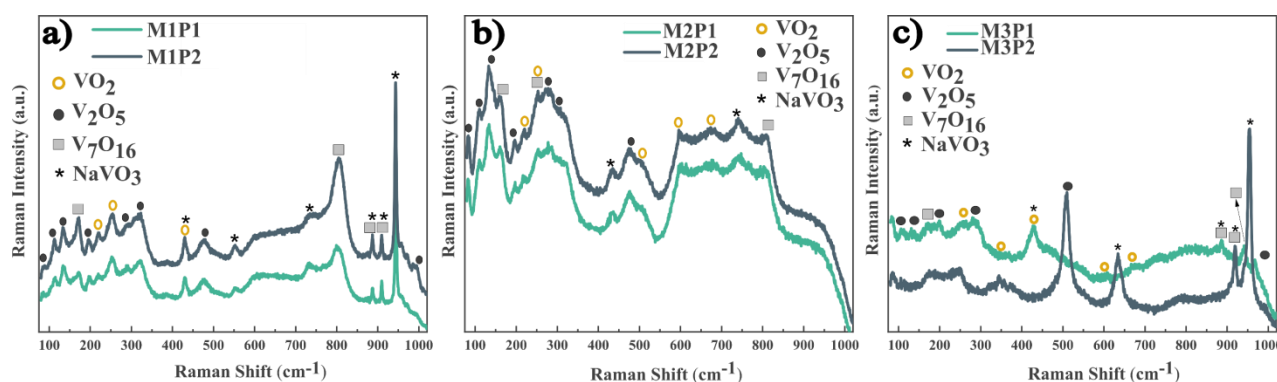


Figure 2. Raman spectra for films deposited at 7.3  $\text{J}/\text{cm}^2$  (a), 13.3  $\text{J}/\text{cm}^2$  (b) and 30.6  $\text{J}/\text{cm}^2$  (c). P1 and P2 refer to the different points in which Raman measurements were performed.

### 3.3. XPS Characterization

Figure 3a shows the high-resolution XPS spectra for the sample deposited at 7.3  $\text{J}/\text{cm}^2$  in the region from 540 to 510 eV, which includes the O1s and the V2p orbitals. Four peaks are observed close to 533, 530, 524, and 517 eV. The first two correspond to the O1s, while the last two correspond to the

V2p<sub>1/2</sub> and V2p<sub>3/2</sub> doublet. For a detailed analysis the peaks were fitted using Voigt profile line shapes. In the O1s region, components at 533.4 and 531.1 eV are assigned to Si–O ( $\text{SiO}_2$ ) and O–V bonds, respectively. In the V2p region, two doublets are present: 524.8 and 517.9, and 523.4 and 516.0 eV, corresponding to  $\text{V}^{5+}$  and  $\text{V}^{4+}$  oxidation states respectively (Moulder *et al.*, 1992; Ureña-Begara *et al.*, 2017).

These results suggest a mixture of  $V^{4+}$  and  $V^{5+}$  composing the film. Additionally, it is observed that the signal corresponding to the  $V^{5+}$  state ( $V_2O_5$ ) is of higher intensity than the  $V^{4+}$  ( $VO_2$ ), indicating a higher amount of  $V_2O_5$  on the surface of this sample. The shape of the  $V2p_{1/2}$  and  $V2p_{3/2}$  signals is bell-shaped with a broad base. It has been reported that the presence of  $V^{4+}$  decreases the slope of the signals corresponding to  $V_2O_5$  (Mendialdua *et al.*, 1995).

In Figure 3b, the XPS high-resolution spectrum of the film prepared at  $13.3 \text{ J/cm}^2$  is presented. The same features are observed with intensity changes: the Si–O component is weaker relative to O–V, and the  $V^{5+}$  signal increases relative to  $V^{4+}$ , indicating a higher  $V_2O_5$  fraction. Notably, vanadium-oxide nanotubes typically contain  $V^{4+}$  and  $V^{5+}$  oxidation states mixed (Souza Filho *et al.*, 2004).

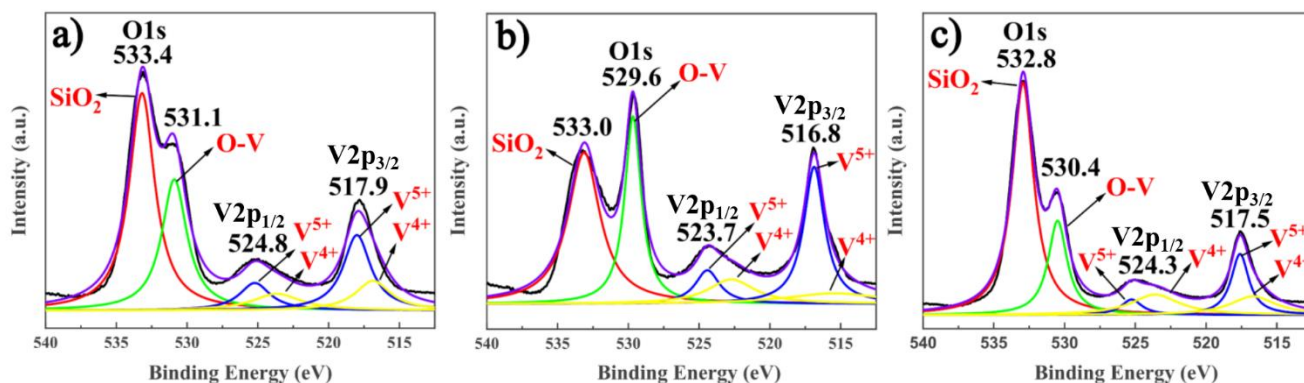


Figure 3. High resolution XPS spectra (O1s and V2p regions) for films deposited at laser fluences of: a)  $7.3 \text{ J/cm}^2$ , b)  $13.3 \text{ J/cm}^2$  and c)  $30.6 \text{ J/cm}^2$ .

For the  $30.6 \text{ J/cm}^2$  film (Figure 3c), the substrate (Si–O) signal is the most intense among the three samples, while the vanadium signals behave similarly to the other spectra. The Si signal likely arises from film discontinuities that expose the glass substrate, consistent with SEM and Raman observations.

### 3.4. Optical properties

Diffuse reflectance UV-Vis spectroscopy was selected because needle-, bar-, and droplet-like morphologies act as scattering centers for incident light (Murphy, 2007). The spectra shown in Figure 4 reveals an absorption edge beginning at approximately  $340 \text{ nm}$  and a broad band from  $\sim 500$  to  $700 \text{ nm}$  consistent with previous reports (Abdullahi *et al.*, 2016). Optical band gaps were estimated using the Kubelka-Munk (K-M) function and Tauc plots (Tauc *et al.*,

1966; Abdullahi *et al.*, 2016). Figure 5 shows Tauc plots for the film deposited at  $30.6 \text{ J/cm}^2$  considering both indirect and direct band transitions. For an indirect transition, two linear regions yield two bandgap values; for a direct transition, only one value is obtained.

The calculated band gap values are shown in Table 1. Direct transitions values ( $\sim 3.2 \text{ eV}$ ) do not match reported values for the phases present in the films. Reported band gaps for  $V_2O_5$  are  $\sim 2.4$  (direct) and  $\sim 2.0 \text{ eV}$  (indirect); for single-walled  $V_2O_5$  nanotubes, the band gap decreases as tube diameter decreases. For semiconducting  $VO_2$  values of  $0.6$ – $0.7 \text{ eV}$  have been reported (Parker *et al.*, 1990; Chain, 1991; Ivanovskaya *et al.*, 2003; Murphy, 2007). When indirect transitions are considered, the obtained values are closer to literature.

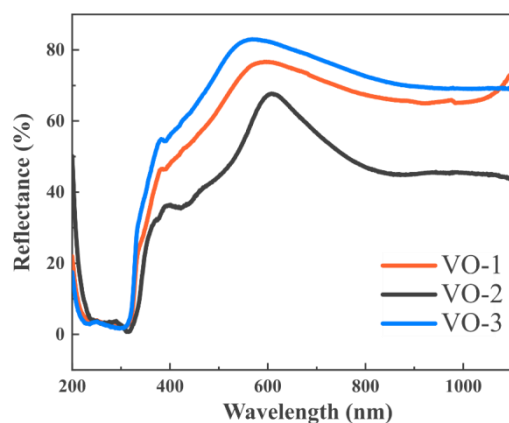


Figure 4. UV-Vis diffuse reflectance spectra for samples deposited at  $7.3 \text{ J/cm}^2$  (VO-1),  $13.3 \text{ J/cm}^2$  (VO-2), and  $30.6 \text{ J/cm}^2$  (VO-3).

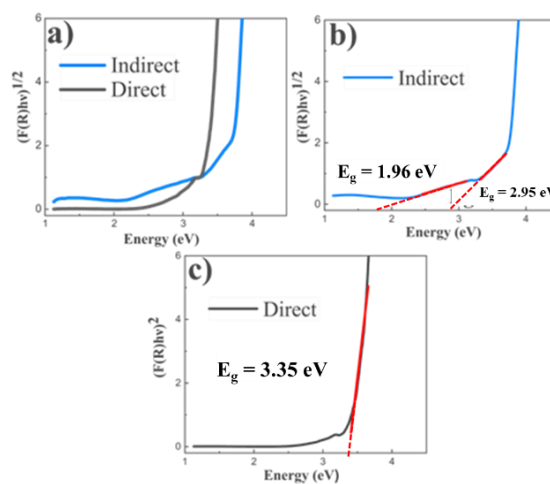


Figure 5. Tauc plots (Kubelka–Munk) for the  $30.6 \text{ J/cm}^2$  film: (a) indirect vs direct comparison; (b) indirect extrapolation; (c) direct extrapolation.



Table 1. Band gap values from diffuse reflectance UV-Vis spectra.

Laser fluence (J/cm <sup>2</sup> )	Band gap (eV)		
	Direct	Indirect	
7.3	3.25	2.91	1.74
13.3	3.20	2.19	1.64
30.6	3.35	2.95	1.96

Figure 6 shows the photoluminescence emission spectra of the deposited films excited with a wavelength of 500 nm. Three emission lines at 544 nm (2.28 eV), 655 nm (1.89 eV) and 730 nm (1.70 eV) for the three films are observed. The 655, and 730 nm emissions can be attributed to split off conduction bands of V<sub>2</sub>O<sub>5</sub> due to its layered structure (Wang et al., 2021).

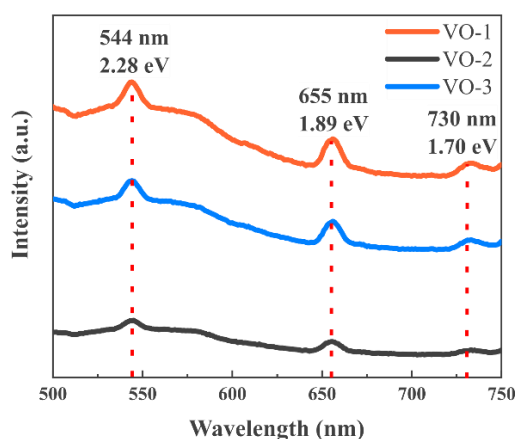


Figure 6. Photoluminescence spectra of the films deposited at 7.3 J/cm<sup>2</sup> (VO-1), 13.3 J/cm<sup>2</sup> (VO-2), and 30.6 J/cm<sup>2</sup> (VO-3).

#### 4. Conclusions

Thin films composed of mixtures of vanadium oxides were obtained by PLD. The multiple oxidation states of vanadium favor the formation of different oxides, including nanotubular structures with mixed oxidation states. Raman and XPS reveal the presence of V<sub>2</sub>O<sub>5</sub>, VO<sub>2</sub> and possibly vanadium oxide nanotubes; NaVO<sub>3</sub>) and SiO<sub>2</sub> were also detected. XPS shows V<sub>2</sub>O<sub>5</sub> as the predominant phase, consistent of the higher stability of V<sup>5+</sup> under these conditions. SEM reveals microneedles and microbars on the film surfaces. The employed approach enables obtaining V<sub>2</sub>O<sub>5</sub> and VO<sub>2</sub> from metallic vanadium without hazardous chemical reagents. These thin film vanadium oxides have potential applications in catalysis, solar cells, smart windows, and optoelectronics.

#### References

Abdullahi, S. et al. (2016) 'Simple Method For The Determination of Band Gap of a Nanopowdered Sample Using Kubelka Munk Theory', *Journal of the Nigerian Association of Mathematical Physics*, 35, pp. 241–246.

Aburas, M. et al. (2019), Thermochromic smart window technologies for building application: A review. *Applied Energy*, 255, p. 113522. <https://doi.org/https://doi.org/10.1016/j.apenergy.2019.113522>.

Aita, C.R. et al. (1986), Optical behavior of sputter-deposited vanadium pentoxide. *Journal of Applied Physics*, 60(2), pp. 749–753. <https://doi.org/10.1063/1.337425>.

Bahlawane, N. and Lenoble, D. (2014), Vanadium Oxide Compounds: Structure, Properties, and Growth from the Gas Phase. *Chemical Vapor Deposition*, 20(7–8–9), pp. 299–311. <https://doi.org/https://doi.org/10.1002/cvde.201400057>.

Bannuru, T. et al. (2008), The electrical and mechanical properties of Au–V and Au–V<sub>2</sub>O<sub>5</sub> thin films for wear-resistant RF MEMS switches. *Journal of Applied Physics*, 103(8), p. 83522. <https://doi.org/10.1063/1.2902954>.

Barceloux, D.G. and Barceloux, D. (1999), Vanadium. *Journal of Toxicology: Clinical Toxicology*, 37(2), pp. 265–278. <https://doi.org/10.1081/CLT-100102425>.

Blum, R.-P. et al. (2007), Surface Metal-Insulator Transition on a Vanadium Pentoxide (001) Single Crystal. *Physical Review Letters*, 99(22), p. 226103. <https://doi.org/10.1103/PhysRevLett.99.226103>.

Chain, E.E. (1991), Optical properties of vanadium dioxide and vanadium pentoxide thin films. *Applied Optics*, 30(19), pp. 2782–2787. <https://doi.org/10.1364/AO.30.002782>.

Chen, C.Y. et al. (2019), Vanadium Oxide as Transparent Carrier-Selective Layer in Silicon Hybrid Solar Cells Promoting Photovoltaic Performances. *ACS Applied Energy Materials*, 2(7), pp. 4873–4881. <https://doi.org/10.1021/acsaem.9b00565>.

Escobar-Alarcón, L., D.A. Solís-Casados, S. Romero, E. Haro-Poniatowski (2024), TiO<sub>2</sub>-Fe<sub>2</sub>O<sub>3</sub> binary thin films prepared by magnetron sputtering for photocatalytic applications. *Materials Science and Engineering B*, 302, 117261.

Escobar-Alarcón, L., E. Camps, M.A. Castro, S. Muhl, J. A. Mejia-Hernandez (2005), Effect of the plasma parameters on the properties of titanium nitride thin films grown by laser ablation. *Applied Physics A*, 81 (6), pp 1221-1226.

Gonzalez-Zavala, F. et al. (2016), Preparation of vanadium oxide thin films modified with Ag using a hybrid deposition configuration. *Applied Physics A: Materials Science and Processing*, 122(4), pp. 1–6. <https://doi.org/10.1007/s00339-016-9991-0>.

Hu, Peng et al. (2023), Vanadium Oxide: Phase Diagrams, Structures, Synthesis, and Applications. *Chemical Reviews*, 123(8), pp. 4353–4415. <https://doi.org/10.1021/acs.chemrev.2c00546>.

Ivanovskaya, V. V et al. (2003), Electronic properties of single-walled V<sub>2</sub>O<sub>5</sub> nanotubes. *Solid State Communications*, 126(9), pp. 489–493. [https://doi.org/https://doi.org/10.1016/S0038-1098\(03\)00254-0](https://doi.org/https://doi.org/10.1016/S0038-1098(03)00254-0).

Julien, C., E. Haro-Poniatowski, L. Escobar-Alarcón, M.A. Camacho-López, J. Jimenez-Jarquín (1999), Growth of V<sub>2</sub>O<sub>5</sub> thin films by pulsed laser deposition and their applications in lithium microbatteries. *Materials Science and Engineering B*, 63, 170-176.

Julien, C., Nazri, G.A. and Bergström, O. (1997), Raman Scattering Studies of Microcrystalline V<sub>6</sub>O<sub>13</sub>. *Physica status solidi (b)*, 201(1), pp. 319–326. [https://doi.org/https://doi.org/10.1002/1521-3951\(199705\)201:1<319::AID-PSSB319>3.0.CO;2-T](https://doi.org/https://doi.org/10.1002/1521-3951(199705)201:1<319::AID-PSSB319>3.0.CO;2-T).

Liu, A. et al. (2007), Vanadium-oxide nanotubes: Synthesis and template-related electrochemical properties. *Electrochemistry Communications*, 9(7), pp. 1766–1771. <https://doi.org/https://doi.org/10.1016/j.elecom.2007.03.027>.

Liu, X. et al. (2006), The effect of thermal annealing and laser irradiation on the microstructure of vanadium oxide nanotubes. *Applied Surface Science*, 253(5), pp. 2747–2751. <https://doi.org/https://doi.org/10.1016/j.apsusc.2006.05.041>.

Mai, L. et al. (2011), Vanadium oxide nanowires for Li-ion batteries. *Journal of Materials Research*. 2011/07/13, 26(17), pp. 2175–2185. <https://doi.org/DOI: 10.1557/jmr.2011.171>.

Mendialdua, J., Casanova, R. and Barbaux, Y. (1995), XPS studies of V<sub>2</sub>O<sub>5</sub>, V<sub>6</sub>O<sub>13</sub>, VO<sub>2</sub> and V<sub>2</sub>O<sub>3</sub>. *Journal of Electron Spectroscopy and Related Phenomena*, 71(3), pp. 249–261. [https://doi.org/https://doi.org/10.1016/0368-2048\(94\)02291-7](https://doi.org/https://doi.org/10.1016/0368-2048(94)02291-7).

Morales Méndez, J. G, B. A. Macías Ayala, A. A. Aguilar Cardoso, J. G. Limas González, L. Escobar Alarcón, M. Picquart, E. Haro Poniatowski (2023), Surface Enhanced Raman Spectroscopy of Methylene Blue Deposited on Ag Nanostructured Substrates prepared by PLD. *Vacuum*, 207: 111580.

Moulder, J.F. et al. (1992), Handbook of X-ray Photoelectron Spectroscopy: A reference book of standard spectra for identification and interpretation of XPS data. Eden Prairie, Minnesota: Physical Electronics Division, Perkin-Elmer Corporation, p. 261.

Muhr, H.-J. et al. (2000), Vanadium Oxide Nanotubes—A New Flexible Vanadate Nanophase. *Advanced Materials*, 12(3), pp. 231–234. [https://doi.org/https://doi.org/10.1002/\(SICI\)1521-4095\(200002\)12:3<231::AID-ADMA231>3.0.CO;2-D](https://doi.org/https://doi.org/10.1002/(SICI)1521-4095(200002)12:3<231::AID-ADMA231>3.0.CO;2-D).

Murphy, A.B. (2007), Band-gap determination from diffuse reflectance measurements of semiconductor films, and application to photoelectrochemical water-splitting. *Solar Energy Materials and Solar*

- Cells, 91(14), pp. 1326–1337. <https://doi.org/10.1016/j.solmat.2007.05.005>.
- Neuville, D.R. (2006), Viscosity, structure and mixing in (Ca, Na) silicate melts. *Chemical Geology*, 229(1), pp. 28–41. <https://doi.org/https://doi.org/10.1016/j.chemgeo.2006.01.008>.
- Pan, M. et al. (2004), Raman study of the phase transition in VO<sub>2</sub> thin films. *Journal of Crystal Growth*, 268(1), pp. 178–183. <https://doi.org/https://doi.org/10.1016/j.jcrysgro.2004.05.005>.
- Parker, J.C. et al. (1990), Optical properties of vanadium pentoxide determined from ellipsometry and band-structure calculations. *Physical Review B*, 42(8), pp. 5289–5293. <https://doi.org/10.1103/PhysRevB.42.5289>.
- Pooseekheaw, P. et al. (2022), Effect of magnetic field on improvement of photocatalytic performance of V<sub>2</sub>O<sub>5</sub>/TiO<sub>2</sub> nanoheterostructure films prepared by sparking method. *Scientific Reports*, 12(1), p. 2298. <https://doi.org/10.1038/s41598-022-05015-2>.
- Salamati, M. et al. (2019), Preparation of TiO<sub>2</sub>@W-VO<sub>2</sub> thermochromic thin film for the application of energy efficient smart windows and energy modeling studies of the produced glass. *Construction and Building Materials*, 218, pp. 477–482. <https://doi.org/10.1016/j.conbuildmat.2019.05.046>.
- Schilbe, P. (2002), Raman scattering in VO<sub>2</sub>. *Physica B: Condensed Matter*, 316–317, pp. 600–602. [https://doi.org/https://doi.org/10.1016/S0921-4526\(02\)00584-7](https://doi.org/https://doi.org/10.1016/S0921-4526(02)00584-7).
- Selvakumar, N. and Barshilia, H.C. (2012), Review of physical vapor deposited (PVD) spectrally selective coatings for mid- and high-temperature solar thermal applications. *Solar Energy Materials and Solar Cells*, 98, pp. 1–23. <https://doi.org/https://doi.org/10.1016/j.solmat.2011.10.028>.
- Souza Filho, A.G. et al. (2004), Raman Spectra in Vanadate Nanotubes Revisited. *Nano Letters*, 4(11), pp. 2099–2104. <https://doi.org/10.1021/nl0488477>.
- Tauc, J., Grigorovici, R. and Vancu, A. (1966), Optical Properties and Electronic Structure of Amorphous Germanium. *Physica statu solidi (b)*, 15(2), pp. 627–637. <https://doi.org/10.1002/pssb.19660150224>.
- Théry, V. et al. (2016), Role of thermal strain in the metal-insulator and structural phase transition of epitaxial VO<sub>2</sub> films. *Physical Review B*, 93(18), p. 184106. <https://doi.org/10.1103/PhysRevB.93.184106>.
- Ureña-Begara, F., Crunteanu, A. and Raskin, J.-P. (2017), Raman and XPS characterization of vanadium oxide thin films with temperature. *Applied Surface Science*, 403, pp. 717–727. <https://doi.org/https://doi.org/10.1016/j.apsusc.2017.01.160>.
- Utegulov, Z.N. et al. (2003), Structural characterization of Eu<sub>2</sub>O<sub>3</sub>–MgO–Na<sub>2</sub>O–Al<sub>2</sub>O<sub>3</sub>–SiO<sub>2</sub> glasses with varying Eu<sub>2</sub>O<sub>3</sub> content: Raman and NMR studies. *Journal of Non-Crystalline Solids*, 315(1), pp. 43–53. [https://doi.org/https://doi.org/10.1016/S0022-3093\(02\)01594-6](https://doi.org/https://doi.org/10.1016/S0022-3093(02)01594-6).
- Wang, C. C. et al. (2021), Structure and Photoluminescence Properties of Thermally Synthesized V<sub>2</sub>O<sub>5</sub> and Al-Doped V<sub>2</sub>O<sub>5</sub> Nanostructures. *Materials* 14(2), 359. <https://doi.org/10.3390/ma14020359>.
- Wang, N. et al. (2016)y, Terbium-Doped VO<sub>2</sub> Thin Films: Reduced Phase Transition Temperature and Largely Enhanced Luminous Transmittance. *Langmuir*, 32(3), pp. 759–764. <https://doi.org/10.1021/acs.langmuir.5b04212>.
- Zhang, Y. et al. (2012), Preparation of W- and Mo-doped VO<sub>2</sub>(M) by ethanol reduction of peroxovanadium complexes and their phase transition and optical switching properties. *Journal of Alloys and Compounds*, 544, pp. 30–36. <https://doi.org/https://doi.org/10.1016/j.jallcom.2012.07.093>.
- Zhou, Q. et al. (2020), Boron doped M-phase VO<sub>2</sub> nanoparticles with low metal-insulator phase transition temperature for smart windows. *Ceramics International*, 46(4), pp. 4786–4794. <https://doi.org/https://doi.org/10.1016/j.ceramint.2019.10.211>.